

**ELECTROCHEMICAL “STRESS” –**  
**COMBINED LOADING IN THE NAVAL ENVIRONMENT**  
PERSPECTIVES ON TREATING ENVIRONMENTAL LOADING  
AT MACRO AND MICRO SCALES

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# Introduction

- Stress-free and stress-assisted corrosion
  - Significant in Naval aircraft structures
- Environmental effects increase the complexity of life prediction models
- Engineering and lifing models needed
  - Account for both mechanical and environmental crack driving forces
  - AND their interactions
    - i.e. covariance / codependence

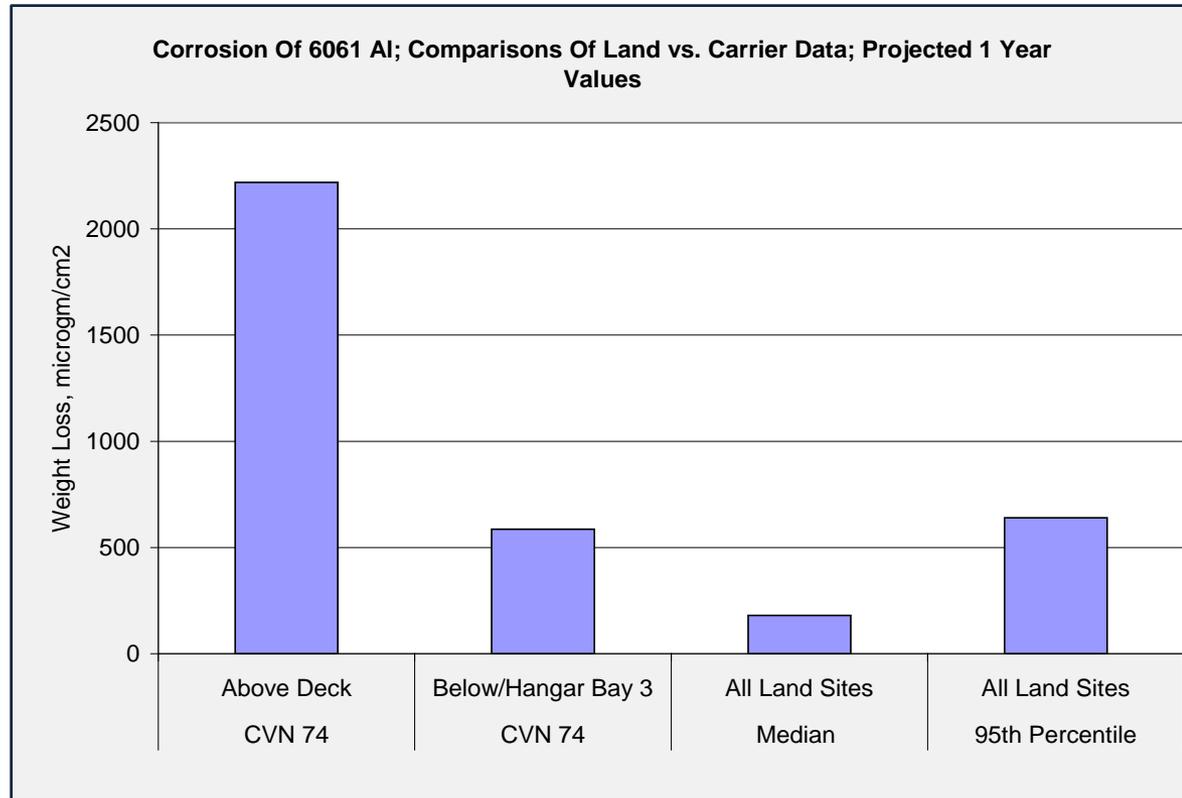
# Experience

- Metallic Structure Durability
  - Joint service, fixed wing tear down analysis
    - Approximately 80% of structural cracks initiated from a corrosion pit
  - Navy FRC structural engineering assessment of tear-down / rework
    - 60/40 split of stress-corrosion cracking / fatigue (mechanical & corrosion)
    - **75-90% of structural damage initiated at galvanic interfaces**
  - OSD Geographic Corrosivity Study
    - Average ship deck environment 4-8X more aggressive than worst land site
  - **Materials selection – primarily mechanical function / ease of purchase**
    - Cannot assess structural risk – location or relative severity of galvanic stress
    - Cannot conduct material alternative analysis for galvanic mitigation
  - **Coatings, Sealants & Polymers – performance varies – T, RH, application**
    - Application, cure & repair – environment and space constraints aboard ship
    - Moisture permeation and galvanic interactions

# Geographic Assessment



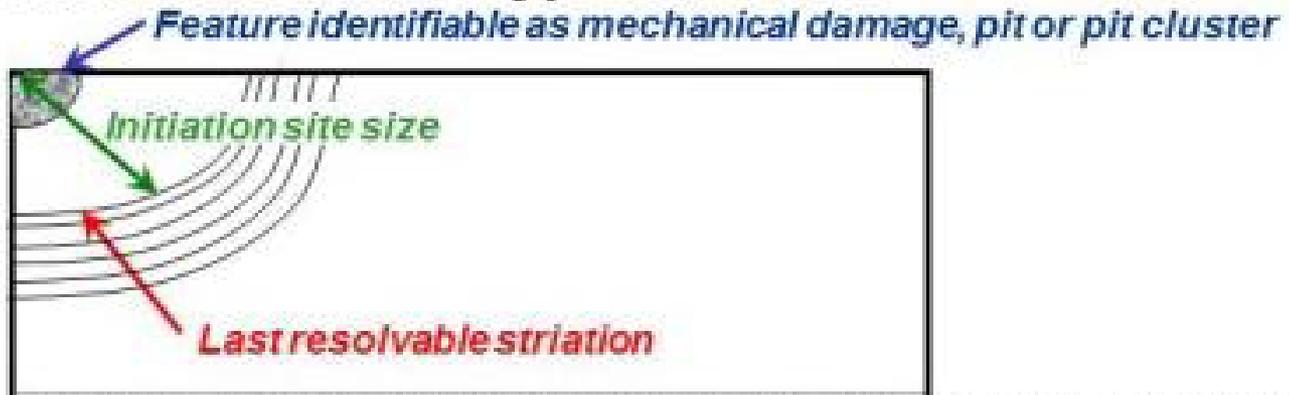
Carrier Exposure Rack on CVN 74



(Abbott and Kinzie, 2004)

# Environmental Structural Damage

- What is “initiation feature” type and size?

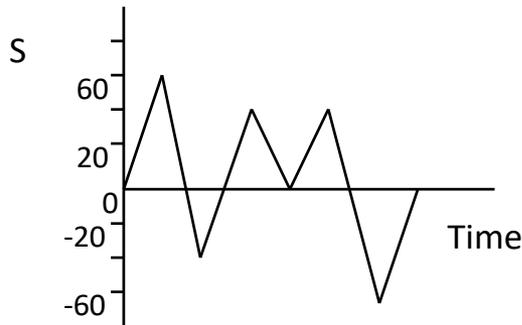


- Even with conservative approach, the largest site is 0.025 in
  - 90% are less than 0.010 in
  - 48% are less than 0.005 in

Initiation Feature	%	Dimensions (in)		
		Minimum	Maximum	Average
Corrosion Pit	80%	0.001	0.025	0.005
Mechanical Damage	20%	0.002	0.013	0.006
Percentage of Initiation Sites on Faying Surface				31%

- Most damage initiated from corrosion pits

# SAFE-Life Approach to Structural Integrity



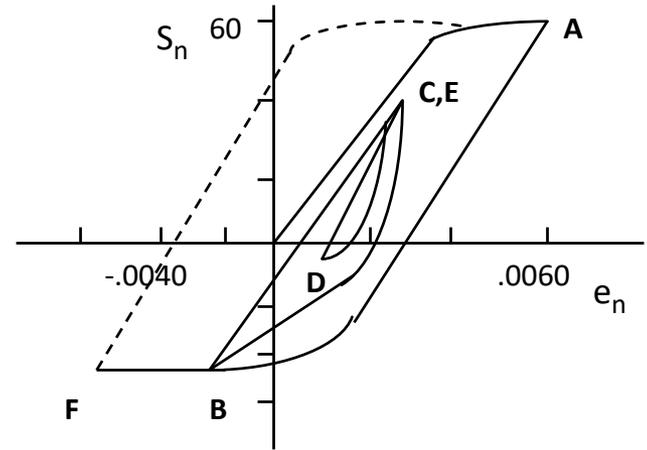
Component Stress History

Neuber's Rule

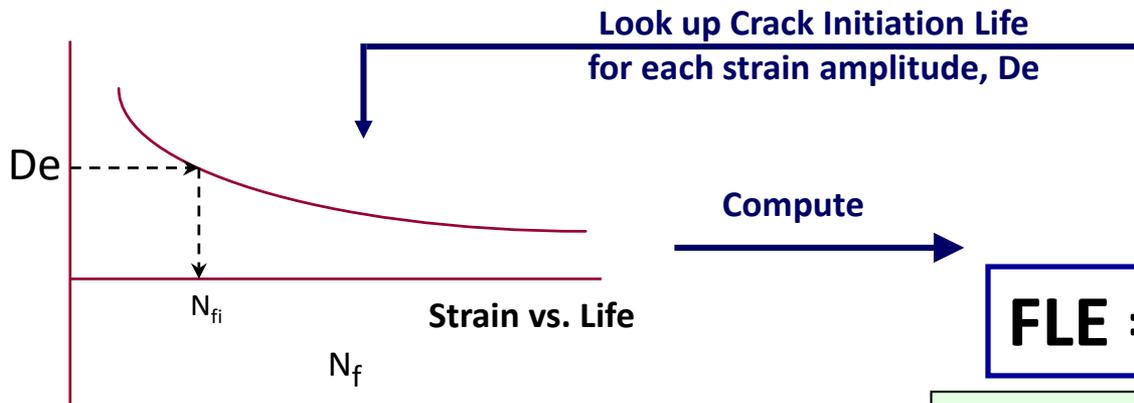
$$K_t = \sqrt{K_\sigma K_\epsilon}$$



Rainflow Cycle Count



Notch Stress-Strain Response



Look up Crack Initiation Life for each strain amplitude,  $D_e$

Compute

$$\text{Damage} = \dot{a} \frac{n_i}{N_{fi}}$$

$$\text{FLE} = 2 * \text{Damage} * 100\%$$

Loads Only Analysis – No Accounting for Environmental Effects

Damage Index Calibrated to FSFT Results (Life to 0.010 in. crack)

# What Makes Corrosion Management of Airframe Designs a Challenge?

## Influence of multiple parameters

- Stress/Strain Level – Plastic Straining
- Strain Rate and Hold Time
- Degree of Electrochemical Activity -Electrolyte, pH, Dissolved O<sub>2</sub>
- Temperature
- Microstructure
- Surface Conditions
- History of environmental loading
- ...

Corrosion effects cannot be separated from structural life predictions  
How to include corrosion considerations during design stage?

# Challenges in Modeling

- Environmental driving forces at a critical location
  - may be significantly more aggressive than those of bulk environments
- Experimental characterization of electrochemistry
  - [potential, pH, concentration, activity] at critical location such as at a crack or defect is difficult
- Surface effects largely ignored
  - diffusion, electrical resistivity, thermal conductivity, magnetic permeability, etc.
- Analytical models are specific to individual mechanisms
  - no model fits all forms of corrosion

# Challenges in Modeling

- Working Hypotheses
  1. Environmental loading analogous to mechanical loading
    - driving Force – electro-chemical, chemical, thermal
    - history / accumulation Effects
  2. “TYPE” of corrosion
    - galvanic coupling is a type of applied stress
    - pitting, anodic dissolution, etc. are types of response, i.e. corrosion/local damage
  3. Galvanic coupling analogous to stress intensity
    - current density distribution – stress intensity
    - reaction/response – strain, dislocation, ionization, dissolution, etc.
  4. Localized damage initiation linked to crack nucleation
    - $D_{\text{local}}$  controlled by surface chemistry and morphology – influenced by mechanical surface energy
    - $D_{\text{nuc}}$  controlled by microstructure and strain response – influenced by surface chemistry
- Note the importance of electrochemical galvanic activity, both at macro and micro scale

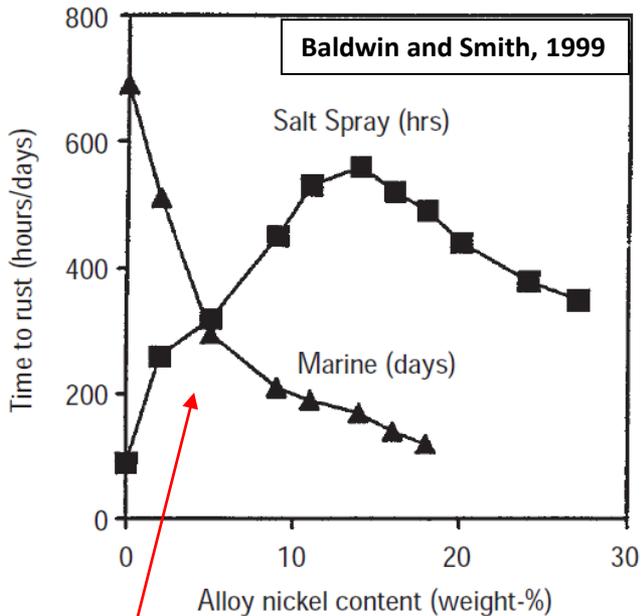
# Issue of Corrosion for Naval Aircraft

Accelerated tests

Vs.

Marine exposure

- The drawbacks with some current accelerated corrosion tests are **poor correlation** to real-world exposure and the **inconsistency** of test results.



In the salt spray test, the corrosion resistance of the zinc-nickel alloy coatings on steel initially increases with increasing nickel content while marine exposure shows opposite trend.

Table I Corrosion rankings for some aerospace aluminium alloys

Marine exposure	Neutral salt spray	Alternate immersion
Clad 7075-T6	Clad 2024-T3	Clad 7075-T6
Clad 2024-T3	Clad 7075-T6	Clad 2024-T3
7075-T6	2024-T3	7075-T6
2014-T3	7075-T6	2024-T3
2014-T6	2014-T3	2014-T3
2024-T3	2014-T6	2014-T6

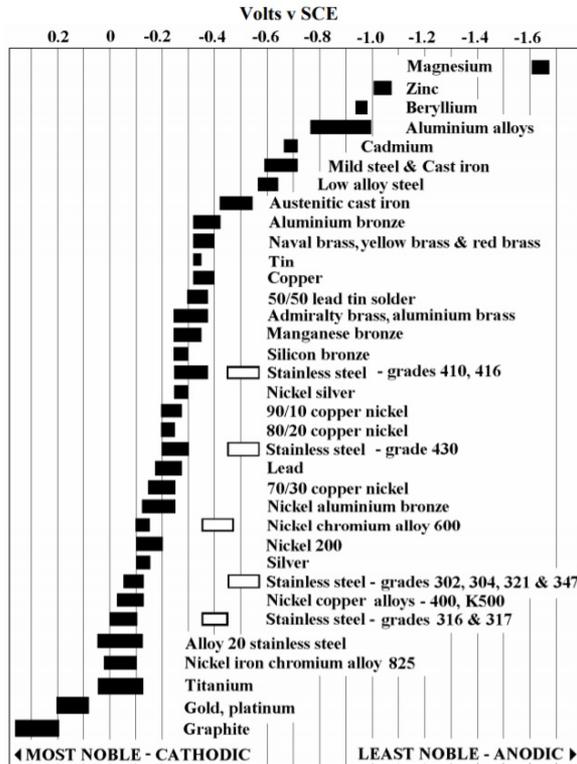
Neutral salt spray test fails to reproduce the ranking orders obtained for any of the materials (above) when compared with marine exposure.

# Issue of Corrosion for Naval Aircraft

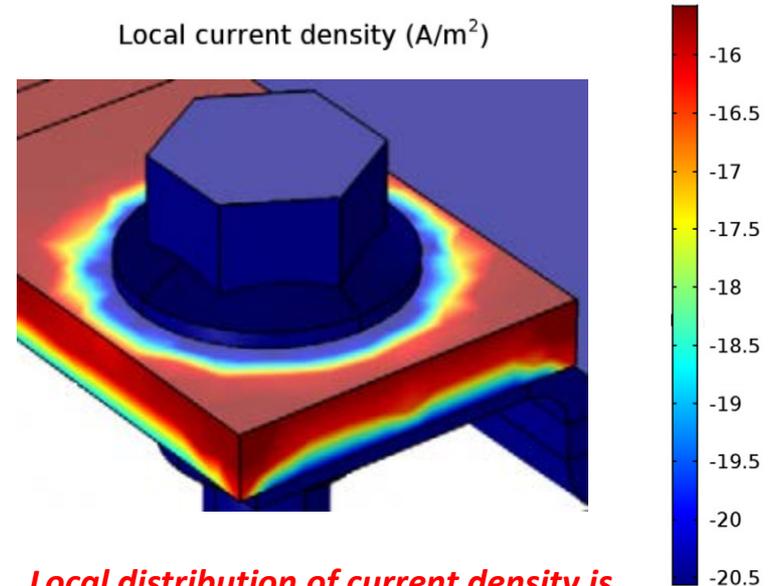
Galvanic series

Vs.

Current Density



Atlas Steels Technical Note No.7



*Local distribution of current density is critical for damage assessment.*

- MIL-STD-889 currently bases the analysis on galvanic potential.
- There is a need to shift from potential-based analysis to current-based galvanic analysis.

# Materials Selection / Corrosion

- ***Traditional galvanic compatibility theory –  $|\Delta E|$*** 
  - Ignores rate and severity control of current density
  - Ignores location effects of distribution
  - Potential only analysis leads to less durable materials selection
    - ignores conductivity / resistivity effects of cathode surface chemistry
    - exchange current density at cathode surface dominates localization
  - Does not account well for thin-film, high O<sub>2</sub> transport conditions
  - Equilibrium vs. actual reaction conditions
    - scalability, extrapolation from OCP often inaccurate
    - geometry and contact area / electrolyte-coverage area effects missing
  - **$\Delta E \geq \Delta E_{th}$  is a more accurate view of galvanic susceptibility**

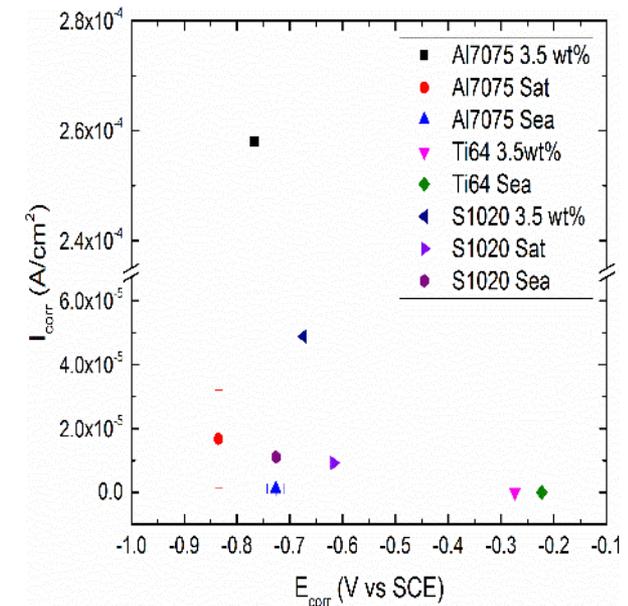
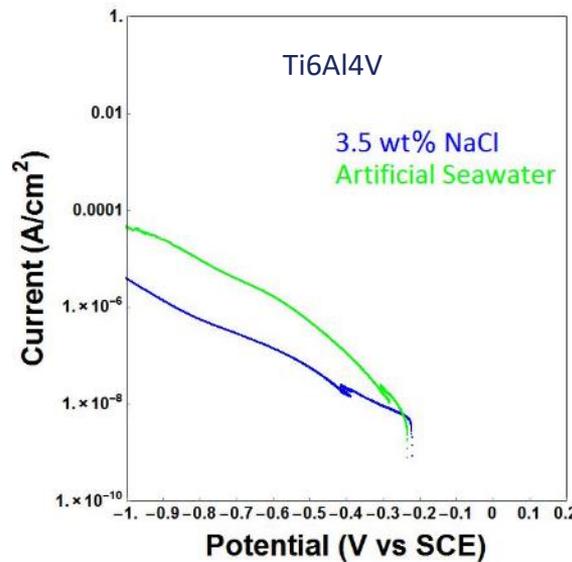
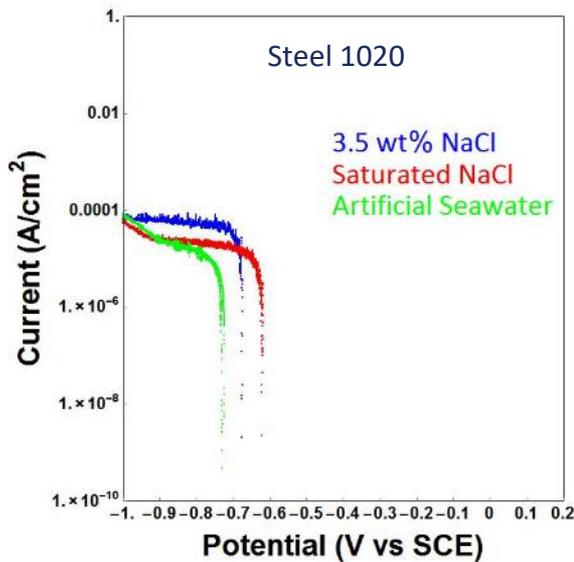
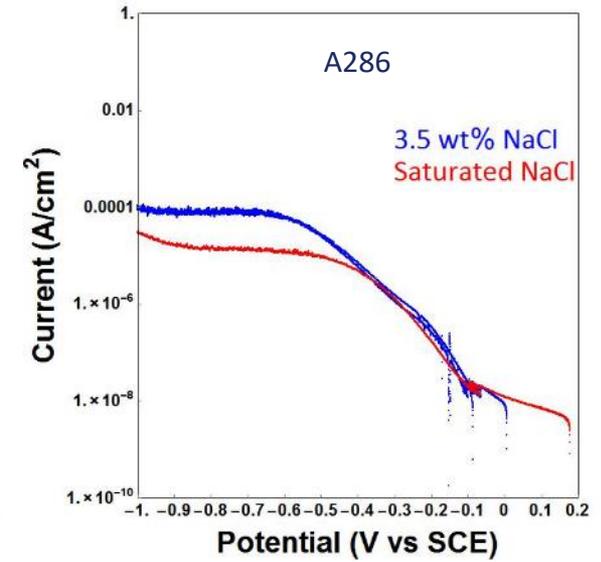
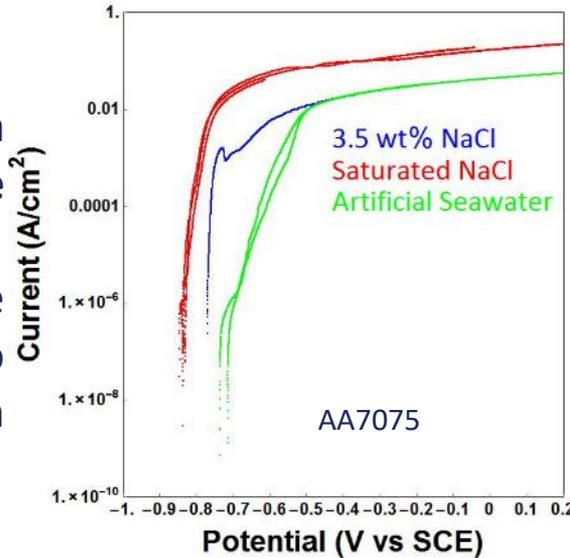
# Galvanic Series

- **Galvanic Series in Seawater:** This well known series indicates the relative nobility of different metals and alloys in seawater, based on the measurement of corrosion potentials. In a galvanic cell, the more noble material in this series will become the cathode (no metal dissolution), while the less noble material will corrode as the anode. **A greater separation of the materials in the galvanic series indicates a bigger potential difference between the materials; generally indicating a greater degree of galvanic incompatibility when coupled.**
- Note that this galvanic series was derived for **one specific electrolyte** (seawater) only. **The materials can have a different nobility ranking in different environments and at different temperatures. The series is based on (averaged) corrosion potential data and therefore does not give a direct indication of the rate of galvanic corrosion.**

# Validation: Laboratory Data

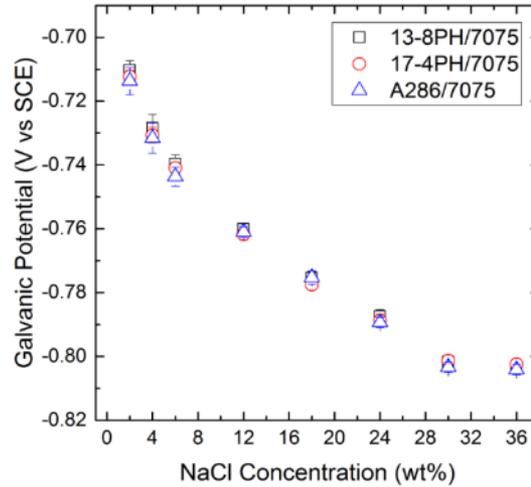
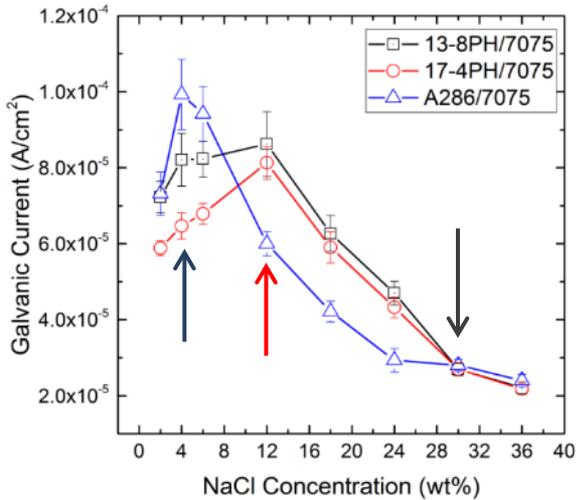
## Polarization Data

- Noticeable differences as a function of electrolyte composition are observed.
- Careful consideration at the time of choosing electrolyte to ensure consistency for both  $E_{\text{corr}}$  and  $I_{\text{corr}}$



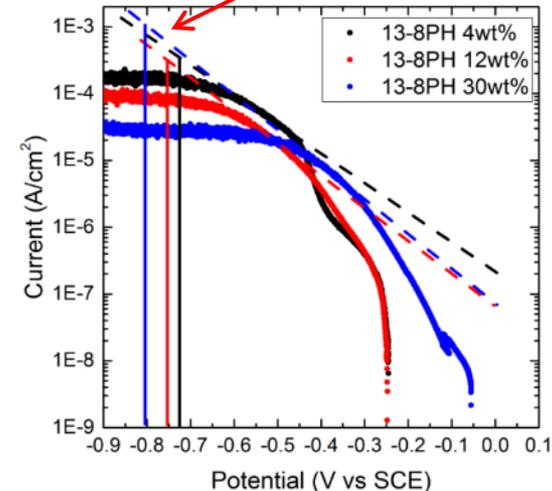
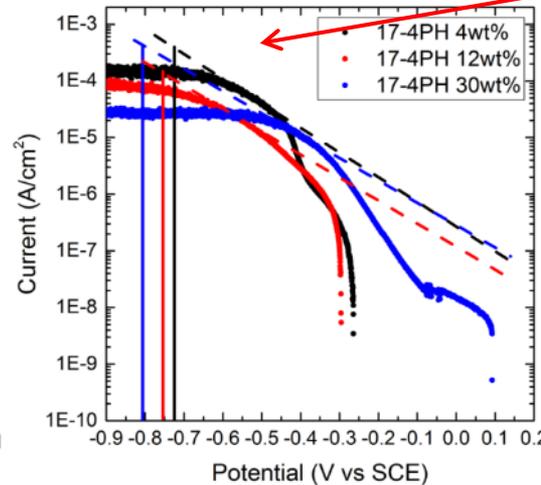
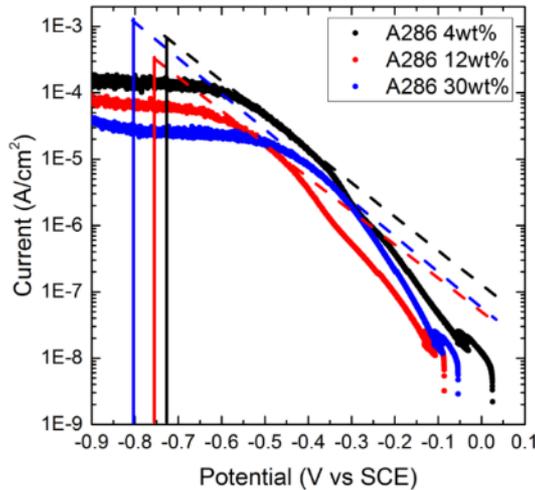
# Validation: Laboratory Data

- For all couples studied, galvanic potential is controlled by Al and current by the cathode material.
- Galvanic current for the different cathode materials peaks at a different concentration.**



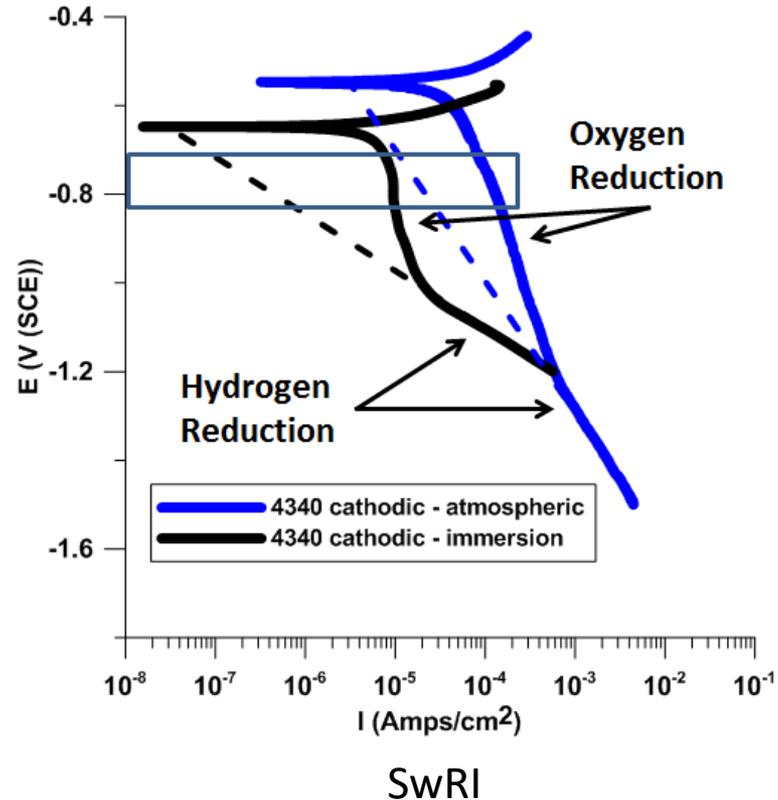
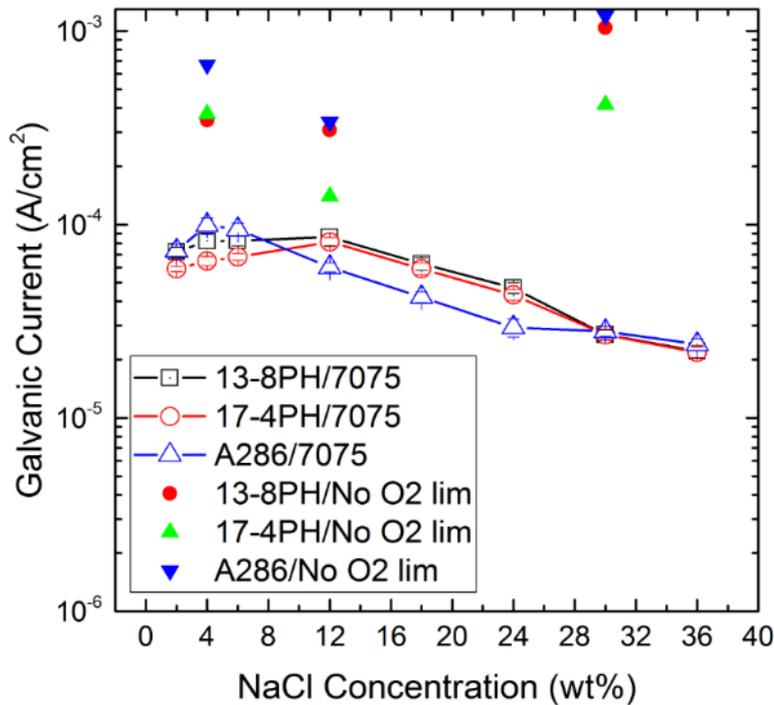
$O_2$  diffusion limitations exacerbated at higher salt concentrations.

Extrapolation of  $O_2$  reduction reaction gives max galvanic current at thin film conditions.



# Validation: Laboratory Data

- Extrapolation to the non-limiting region is confirmed by non-immersion measurements on steel 4340.



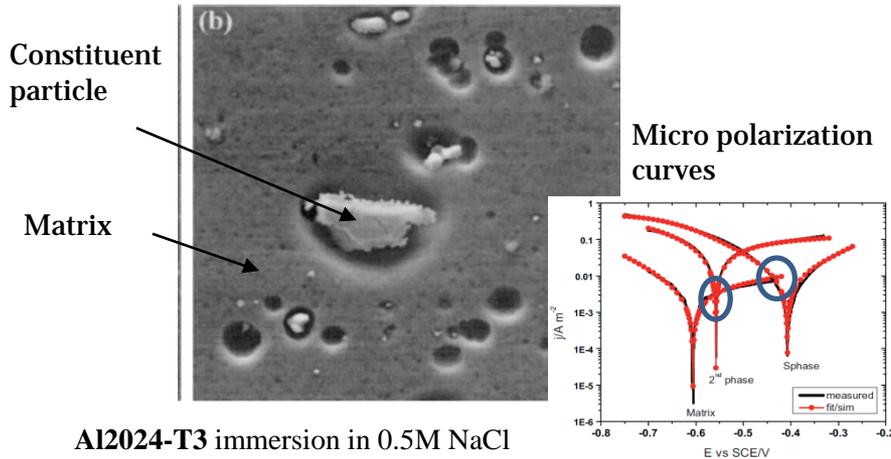
**Kinetics of O<sub>2</sub> reduction reaction (cathode) control galvanic current!**

# Galvanic Activity, at Macro and Micro Scales

- Macroscopic galvanic corrosion is a major factor in Naval structures
- Stress assisted microscopic galvanic corrosion is significant in aircraft aluminum alloys (2xxx and 7xxx series aluminum)

# Galvanic Activity at Macro and Micro Scales

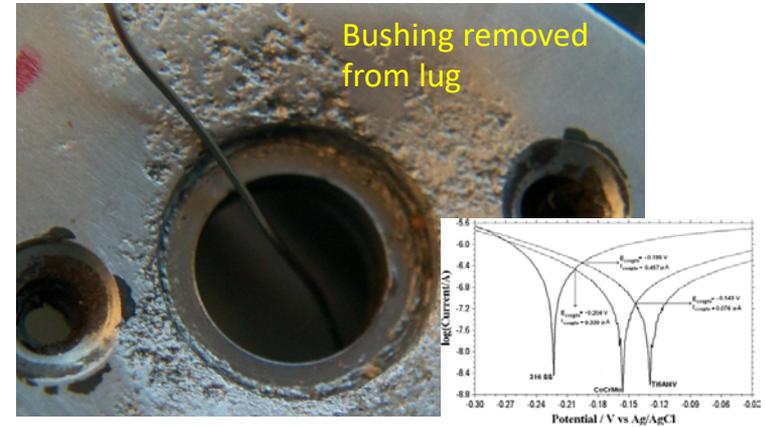
## Micro-galvanic corrosion



Al2024-T3 immersion in 0.5M NaCl

Source: R.P. Wei, et al., *Metall. Mater. Trans A*, 39, 486-492, 2002.

## Macro-galvanic corrosion



### Micro-galvanic cell between constituent particles

- Provide local corrosion activity
- Provide microscale interaction between stress and corrosion
- Develop a multi-physics program for environmentally assisted crack modeling
- Provide guidance for microstructural optimization

### Macro-galvanic cell between dissimilar metals

- Provide corrosion hot spot
- Provide macro galvanic corrosion current
- Provide driving force for corrosion damage
- Develop a numerical tool to predict galvanic corrosion rate
- Provide design specifications

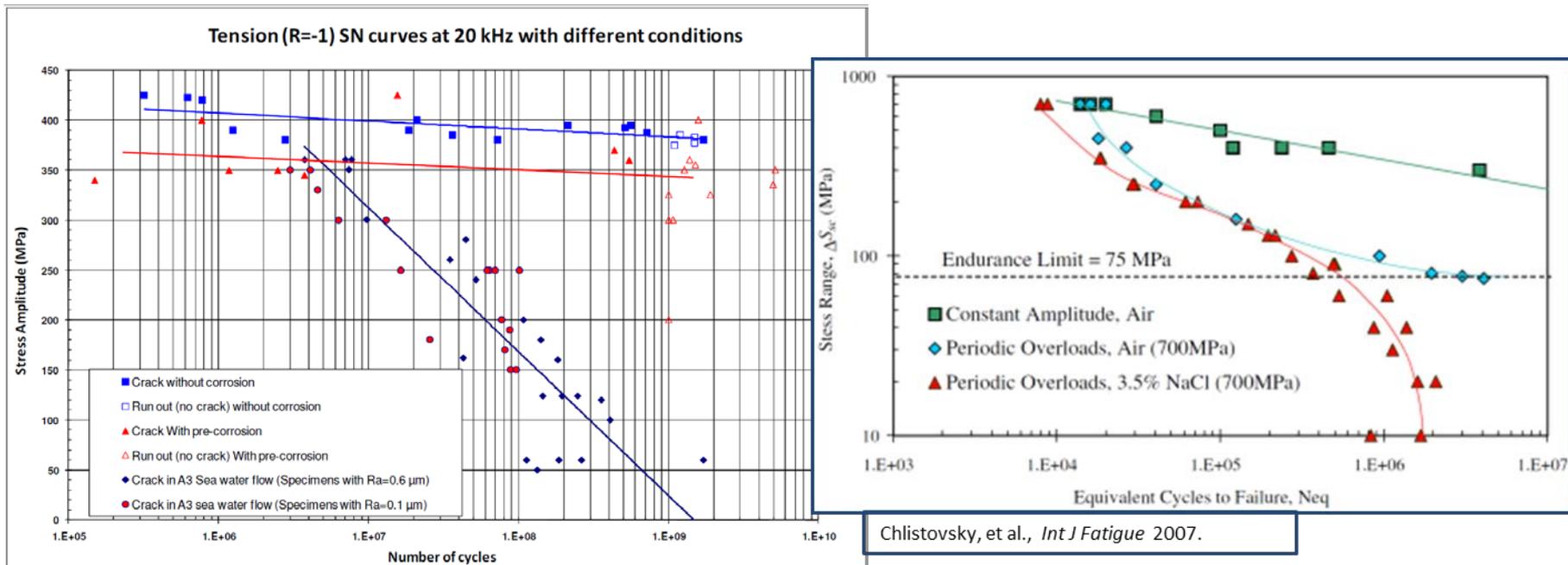
# Issue of Corrosion for Naval Aircraft

Pre-corrosion fatigue

Vs.

Corrosion fatigue

- Evidence of synergistic effect between **fatigue** and **corrosion**.
- The resulting damage is **higher** compared to when each is active individually.



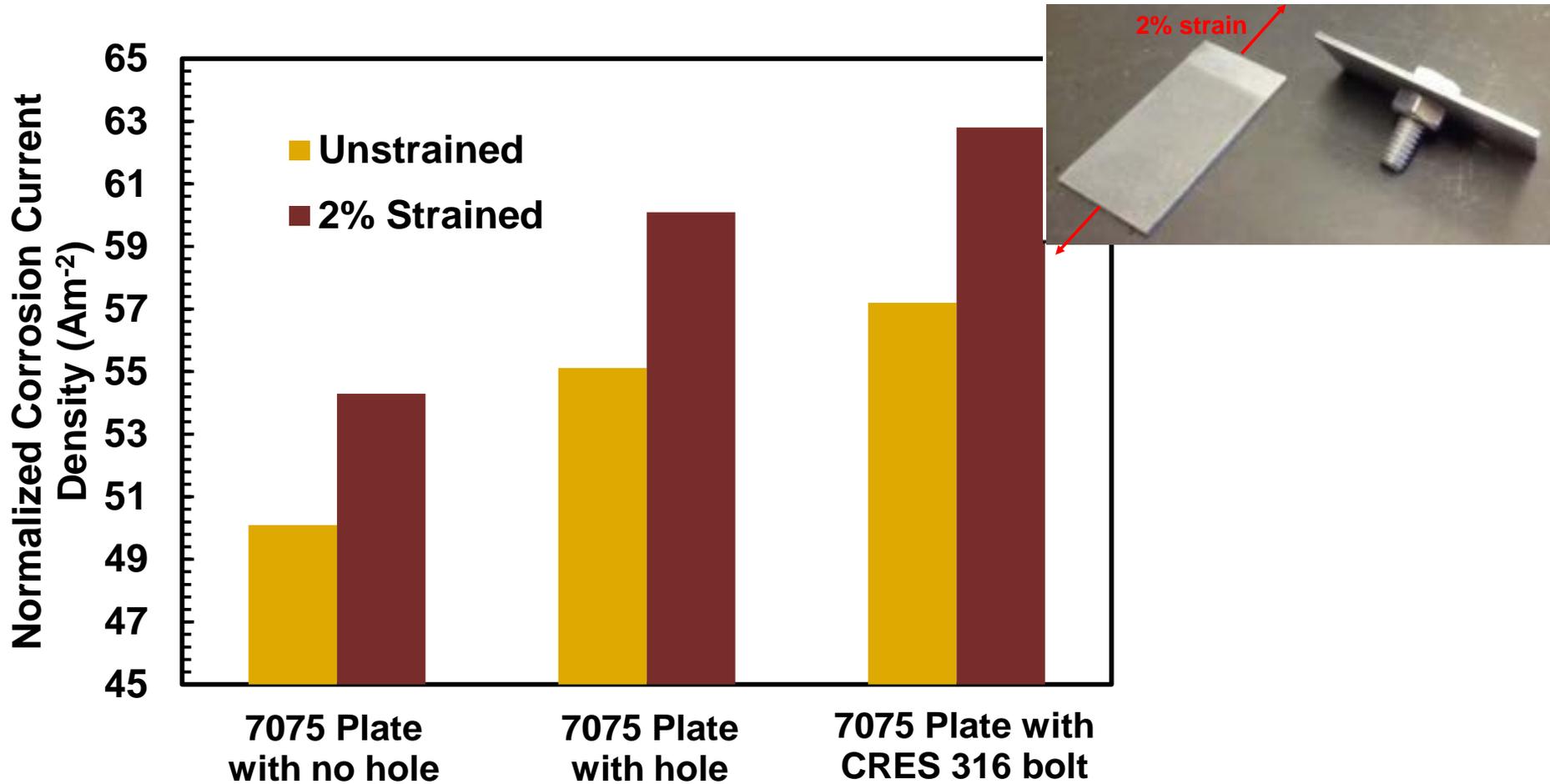
The fatigue strength at  $10^8$  cycles is significantly reduced by a factor of 74% compared to the virgin specimens and of 71% compared to the pre-corroded ones. **Dramatic reduction in strength** is observed due to simultaneous corrosion-fatigue in the UHCF regime.

# Effect of Mechanical Load/Stress

- Increased electrochemical activity
  - Mechanical disruption of an otherwise protective or metastable corrosion oxide film
  - Inhibition of surface oxide formation due to hydrolytic acidification of the crack tip electrolyte
  - Extrusion of high index slip planes that dissolve at a larger rate than the original low index planes at a metal surface
  - Increased anodic dissolution current density due to cathodic shift of the corrosion potential

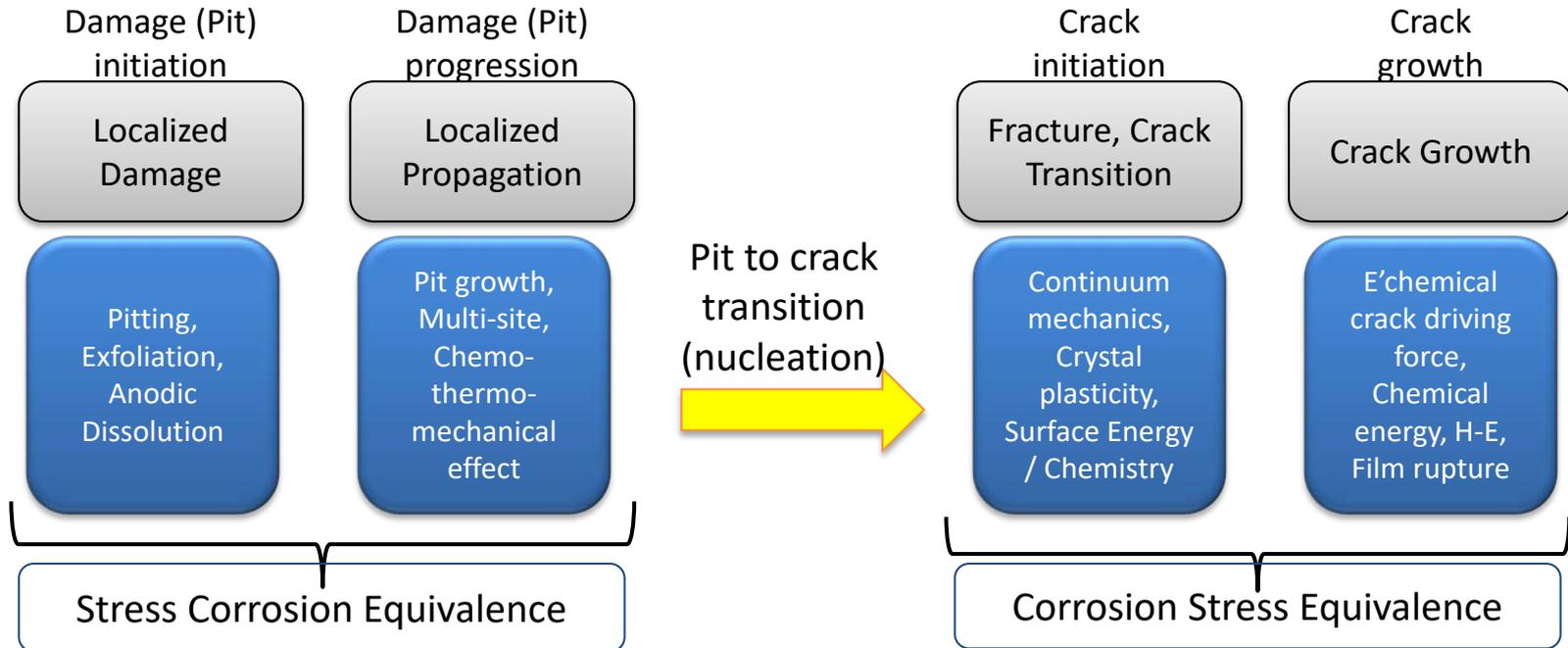
Reliable quantification of effects is an on-going process with  
**continuous model verification and validation**

# Effect of Load on Corrosion Current

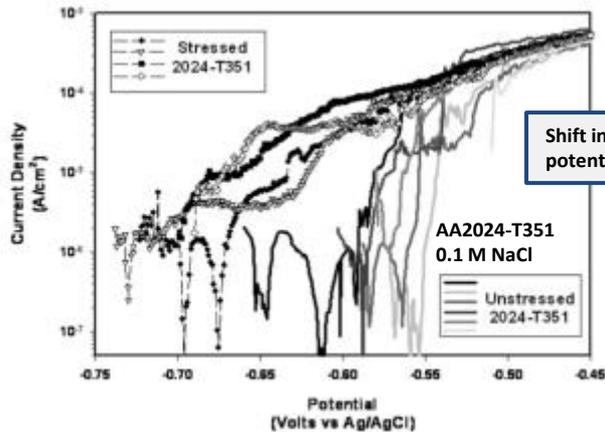


K. Solanki, ASU [ONR Grant] Experimentally obtained corrosion current for Al 7075 plate for strained and unstrained sample.

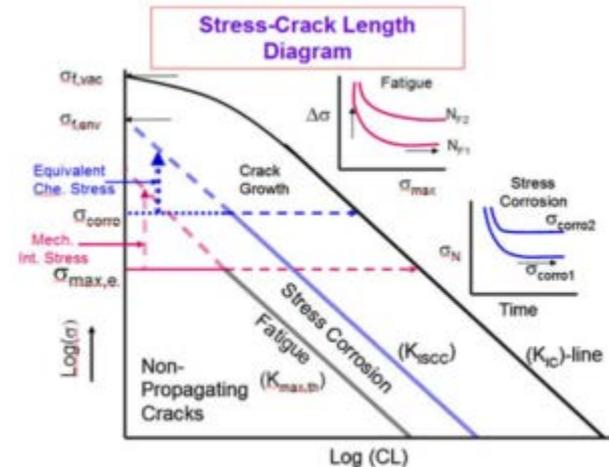
# Corrosion stress equivalence



$$[I/A] J_{corr} \propto \sigma [F/A]$$



$$[I^2x/position] K_{corr} \propto K_{(\Delta K, K_{max}, K_{\perp})}$$



# Corrosion Crack Initiation and Growth

## Environmental Crack Driving Force (ECDF)

### Objective

Develop analytical expressions relating electrochemical crack driving force (ECDF) to solution conductivity, pH, anion and cation concentrations, aqueous hydrogen concentration and temperature.

### Mechanical Crack Driving Force:

$$G(\text{joules/m}^2) = -\left(\frac{dU_e}{dA}\right)_\Delta = -\frac{1}{B}\left(\frac{dU_e}{da}\right)_\Delta$$

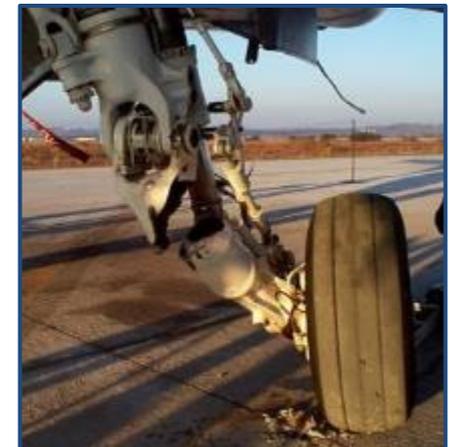
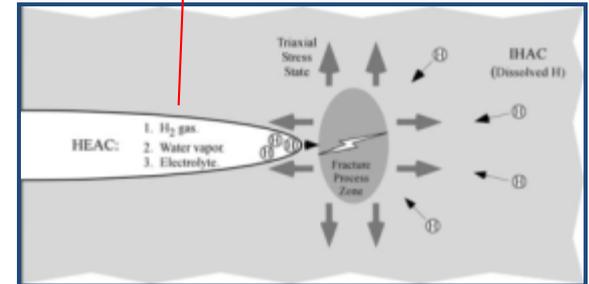
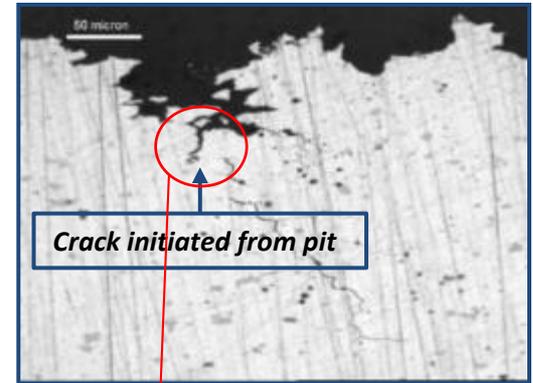
### Electrochemical Crack Driving Force:

1: Hydrogen Embrittlement:

$$\tilde{G}_c(\text{joules/m}^2) = -\left(\frac{d\Delta\tilde{G}_H}{dA}\right)_{T,P} = \left(\delta_{int}\hat{\theta}_{int}/2\bar{V}_M\right) \times \left[zF\eta_c - \frac{RT}{2}\ln(f_{H_2(a)})\right]$$

2: Active Path Dissolution :

$$\tilde{G}_a(\text{joules/m}^2) = -\frac{d\Delta G_a}{dA} = (\delta_{cor}/2\bar{V}_m)zF\eta_a$$



F/A-18 lever arm collapse due to corrosion+sustained stress

$$\left(\frac{da}{dN}\right)_{CF} = \Phi \left(\frac{da}{dN}\right)_{inert} + \left(\frac{da}{dN}\right)_{EAC}$$

Environmentally Assisted Crack Growth

# Issue of Corrosion for Naval Aircraft

Global Env.

Vs.

Local Env.

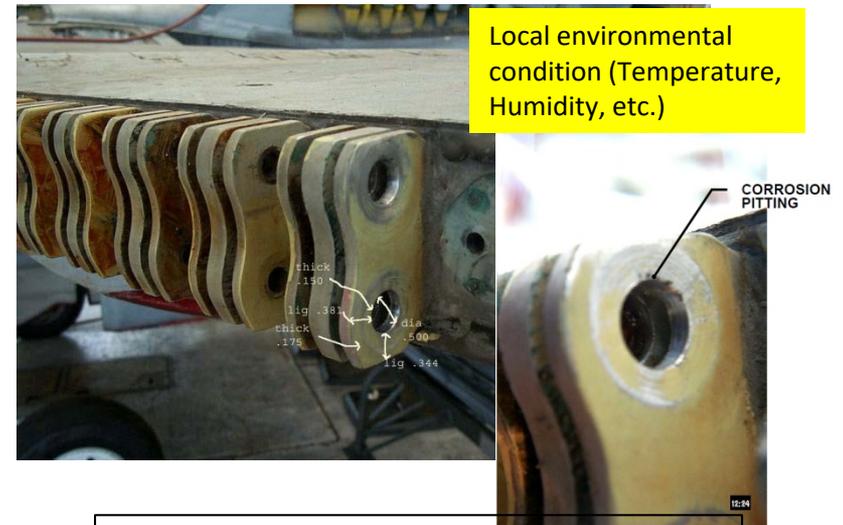
- **Local** environment (part scale) can be significantly different from **global** environment (system scale)

Global environmental condition (Temperature, Humidity, etc.)



- **Global** weather conditions
- Squadron location
- Shelter condition
- Flight requirement
- Wash cycle
- Ground activity

Local environmental condition (Temperature, Humidity, etc.)

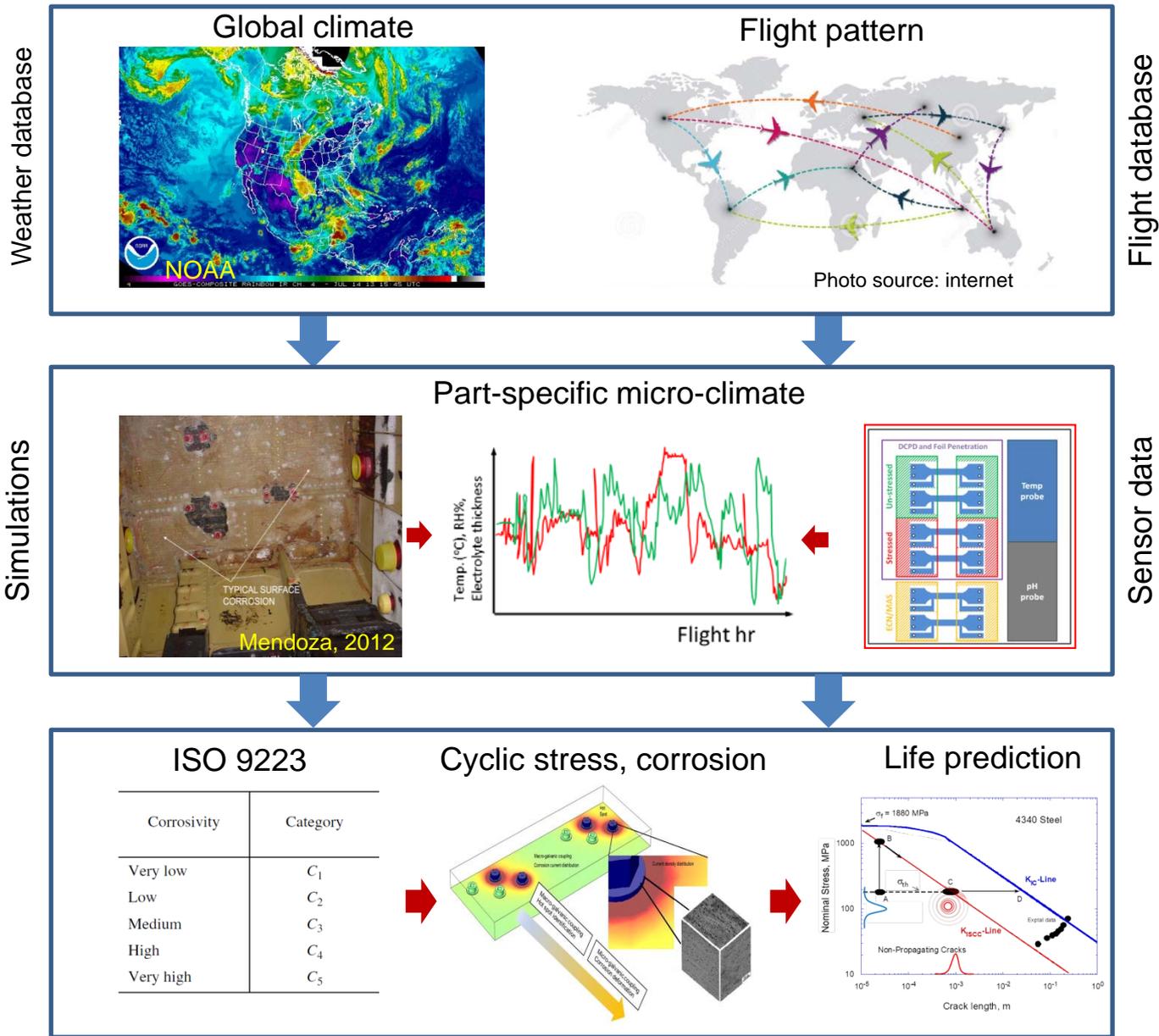


- **Local** environment
- Dissimilar materials
- Part geometry, lap joints, fasteners
- Capillary action
- Occluded areas
- Accumulation of dirt and debris
- Coatings
- Mechanical loads

# The Need for an Environmental Builder

- **Incorporate available **sensor data****
  - Sampled or continuous information at sensor location(s) on aircraft
- **Link the ground station and sensor information to predict **local/part environment****
  - Need a transfer function
- **Extract damaging events from assembling the dynamic environment - **environment history****
  - Identify type of corrosion type and extent at different locations
- **Invoke suitable prediction models by combining '**electrochemical stress**' with **mechanical stresses****
  - Sensor informed degradation prediction and prognosis
  - Use advanced machine learning techniques to gap fill sparse service data

# Overview of the Environmental Builder



# Prospects for Future Work

- Include the environmental builder as well as life prediction models
- Integrate corrosion modules with structural modules for *incubation life and initiation*
- Combine synergistic actions at micro, meso, and macro levels at occluded regions
- Provide guidance for design data extrapolation
- Make a connection between material design and structural application (Corrosion MMPDS)